

Crucial role of orbital structure in formation of frustrated magnetic structure in BiMnO₃L. E. Gonchar^{1,2,*} and A. E. Nikiforov²¹*Ural State University of Railway Transport, 66 Kolmogorov St., Ekaterinburg 620034, Russia*²*Ural Federal University, 19 Mira St., Ekaterinburg 620002, Russia*

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The paper presents an investigation in the field of orbital physics of strongly correlated oxides. The theoretical study of vibronic mechanism of orbital and magnetic structures forming in BiMnO₃ crystal is carried out. An effect of orbital structure upon superexchange interaction is described. Nonlinear and second-neighbor terms in vibronic interaction on manganese ions play an important role in magnetic ordering of frustrated BiMnO₃. It is shown that the linear vibronic interaction is insufficient to describe the experimentally detected ferromagnetic structure of bismuth manganite. The new approach to orbital structure formation, presented in the paper, could be used not only in manganite physics but also in other Jahn-Teller compounds.

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I. INTRODUCTION

BiMnO₃ crystal is a vivid example of renewal of interest in manganite crystals. The manganite crystals $RMnO_3$ ($R = \text{La, Nd, Pr, Dy, Sm, Tb, Gd, Ho, Y}$) are parent compounds to colossal magnetoresistance crystals and have a peculiar interdependence between charge, lattice, orbital, and magnetic degrees of freedom. A lot of investigations were devoted to the multiferroic (MF) manganites,¹ a combination of magnetic ordering with ferroelectricity (FE). The compound without orbital degeneracy BiFeO₃ is also a good MF crystal.² The mechanisms of FE in these crystals ($RMnO_3$ and BiFeO₃) are different.^{3,4} In manganites, there is a spiral spin structure on manganese sublattice and in the bismuth oxide, a lone pair of electrons of Bi³⁺ ion creates the dipole moment. The MF behavior of ferromagnetic BiMnO₃ was discovered in films⁵ and polycrystals,⁶ but in a single crystal, the FE is impossible due to symmetry considerations.^{4,7} The crystal structure of BiMnO₃ with $C2$ (No. 5) space symmetry is determined in earlier investigations of Refs. 8 and 9. This symmetry group is noncentrosymmetric and allows the coexistence of ferromagnetism (FM) and FE. In Refs. 10 and 11, the later experimental studies of this compound disclaim the noncentrosymmetric space group of this crystal structure and describe the structure by the centrosymmetric space group $C2/c$ (No. 15). This symmetry allows no coexistence of FE and FM in the crystal. The detailed experimental comparison of both symmetries was made in Ref. 12. The conclusion of $C2/c$ symmetry was made by the authors. The FM magnetic ordering is reliably determined by experiments,^{10,11,13,14} thus the FE of the single crystal is not ascertained. The Curie temperature of BiMnO₃ is $T_C \sim 100$ K.^{6,8-13} The temperature of possible FE phase transition is $T_{FE} = 450$ K (Refs. 6 and 15) or 770 K (Refs. 13 and 15). The last value is a result of indirect measurements and is rejected in later investigations.¹² Theoretical and experimental investigations of orbital and magnetic orderings are carried out in order to find if it is possible to switch this crystal into the MF state.

The orbital ordering (OO) of BiMnO₃, estimated by lattice distortion, is not the same type with the OO in LaMnO₃ manganites.^{7,8,11-15} It is formed by interaction of an orbitally degenerated manganese ions sublattice and a

strongly polarized bismuth ions sublattice.¹⁶ The temperature of OO is proposed to be $T_{OO} = 750$ K (Refs. 10 and 15) or 450 K (Refs. 13 and 15). The orbital structure of oxygen octahedra is complicated due to Jahn-Teller distortion. Within each pseudoperovskite direction, the e_g orbital sequence is $d_{3x^2-r^2}/d_{3y^2-r^2}/d_{3x^2-r^2}/d_{3z^2-r^2} \dots$ ^{8-12,15} The orbital structure changes with the crystal structure, which undergoes some phase transition with temperature and external pressure,^{6-14,17} thus, the Jahn-Teller distortions of oxygen octahedra are suppressed by growing temperature and pressure. The theoretical calculation of the orbital structure was carried out in Ref. 18 with the same result: the directions of d orbitals coincide with the elongated Mn–O bonds. Thus the theoretical studies of BiMnO₃ orbital structure were carried out only as analysis of experimentally determined Jahn-Teller distortions of nearest-neighbor oxygen coordination.

The description of the magnetic structure due to OO is based generally on Goodenough-Kanamory rules.⁸⁻¹² The *ab initio* calculations of exchange interaction were made by Soloviev *et al.* in Refs. 7, 16, and 18. In order to find the possibility of FE, the authors propose a model of a complicated canted magnetic structure. Nevertheless, this ordering is inconsistent with experiments confirming collinear FM structure. Multiple magnetic transitions are found under external pressure.^{11,14,19} There is no detailed theoretical analysis of the mechanism of ferromagnetic to antiferromagnetic (AFM) transition of BiMnO₃ under pressure.

Like rare-earth manganites, BiMnO₃ changes crystal, orbital, and magnetic properties with doping. The nonstoichiometry by oxygen²⁰ leads to crystal symmetry changes, including $C2$ space group, FM and AFM structures. There is an ongoing discussion of FE possibility in the nonstoichiometric samples. Other kinds of doping used by the authors are rare-earth and alkaline-earth dopings like in manganites.²¹ Similar to lanthanum-manganite-based compounds, the charge ordered states, metallic behavior, and AFM structures are detected in these investigations.

The effect of additional interactions in the vibronic Hamiltonian upon orbital structures was proposed in Ref. 22 for layered manganites and in Ref. 23 for titanates. These investigations demonstrate the significance of a thorough description of orbital structure and the necessity to advance

the orbital structure model presented earlier for manganites in Refs. 24 and 25.

The aim of this paper is to present an advanced model of orbital structure in order to describe a vibronic mechanism of magnetic structure forming in the case of magnetic frustration. Taking into account the linear vibronic interaction of orbitally degenerated manganese subsystem not only with nearest-neighbor oxygen coordination, but with next-nearest neighboring bismuth environment and additional nonlinear vibronic terms corresponding to nearest neighbors, the FM ordering of bismuth manganite is described. The FM state is a result of orbital degeneration of manganese sublattice and thus is impossible for BiFeO₃, which is a *G*-type AFM crystal.² The FM ordering combined with an FE state promises a wide range of technical applications of the crystal. In spite of the prohibition of this combination of properties for bulk BiMnO₃ crystal, the investigation of the bulk crystal plays an important role in the description of thin-film orbital and magnetic structures.

II. VIBRONIC INTERACTIONS AND ORBITAL STRUCTURE

The crystal structure of BiMnO₃ is determined in the experimental studies of Refs. 10 and 11, and it belongs to the *C2/c* space symmetry group. The lattice constants and crystal structure angles are, correspondingly, $a = 9.5473 \text{ \AA}$, $b = 5.6167 \text{ \AA}$, $c = 9.8699 \text{ \AA}$, $\alpha = \gamma = 90^\circ$, and $\beta = 110.66^\circ$. The crystal structure parameters are shown in Table I.

The monoclinic coordinate system in the crystal is of non-Cartesian type. That is why a pseudoperovskite reference frame is used throughout the paper. The correlation between coordinate axes of different referent frames is described in detail in Appendix.

The crystal structure is of distorted perovskite type. In BiMnO₃ crystal, Mn³⁺ ions have approximately octahedral oxygen coordination. Thus the ground state of this ion is a double-degenerated ⁵*E* state. The degeneration could mainly be removed due to cooperative Jahn-Teller effect as in lanthanum manganite,²⁴ but the strong distortion of the crystal structure may affect the orbital structure as well.^{22,23} A wave function of the ground state of each manganese ion n is a linear combination of eigenfunctions $|\theta\rangle_n$, $|\varepsilon\rangle_n$ of double-degenerated ⁵*E* ground state, used in Ref. 25:

$$\psi_n = \sin \frac{\Phi_n}{2} |\theta\rangle_n + \cos \frac{\Phi_n}{2} |\varepsilon\rangle_n, \quad (1)$$

where Φ_n is a mixing angle of orbital functions.

TABLE I. The *C2/c* crystal structure parameters in monoclinic reference frame according to Ref. 11.

Ion type	<i>x</i>	<i>y</i>	<i>z</i>
Bi (8 <i>f</i>)	0.1375	0.2114	0.1269
Mn1 (4 <i>e</i>)	0.0	0.219	0.75
Mn2 (4 <i>d</i>)	0.25	0.25	0.5
O1 (8 <i>f</i>)	0.097	0.173	0.578
O2 (8 <i>f</i>)	0.145	0.568	0.366
O3 (8 <i>f</i>)	0.356	0.547	0.166

The ground state on Mn³⁺ ion is described by the many-electron configuration $[(t_{2g}^3)^4 A_2, e_g]^5 E$. Thus the many-body eigenfunctions of the ground state are determined as d^4 Slater determinants, depending of single-electron $\tilde{\xi}$, $\tilde{\eta}$, $\tilde{\zeta}$ t_{2g} states and $\tilde{\theta}$, $\tilde{\varepsilon}$ e_g states:

$$|\theta\rangle_n = -|\tilde{\xi} \tilde{\eta} \tilde{\zeta} \tilde{\varepsilon}\rangle_n, \quad |\varepsilon\rangle_n = |\tilde{\xi} \tilde{\eta} \tilde{\zeta} \tilde{\theta}\rangle_n. \quad (2)$$

The corresponding wave function (1) of the orbital ground state may be rewritten as

$$\psi_n = |\tilde{\xi} \tilde{\eta} \tilde{\zeta} \chi\rangle_n, \quad (3)$$

where the single-electron wave function χ_n depends upon the eigenfunction of n th e_g electron $|\tilde{\theta}\rangle_n$, $|\tilde{\varepsilon}\rangle_n$

$$\chi_n = \cos \frac{\Phi_n}{2} |\tilde{\theta}\rangle_n - \sin \frac{\Phi_n}{2} |\tilde{\varepsilon}\rangle_n. \quad (4)$$

This way of describing wave functions is used in many papers, for example, in Ref. 22.

The orbital structure of manganese sublattice is formed by vibronic interactions, which consist of three parts:

$$H_{\text{vib}} = H_{\text{lin}} + H_{QQ} + H_R. \quad (5)$$

The main part of vibronic interaction is the linear electron-lattice interaction due to cooperative Jahn-Teller effect:

$$H_{\text{lin}} = V_e \sum_n (Q_{\theta n} X_{\theta n} + Q_{\varepsilon n} X_{\varepsilon n}), \quad (6)$$

where

$$X_{\theta n} = \begin{pmatrix} -1 & 0 \\ 0 & 1 \end{pmatrix}, \quad X_{\varepsilon n} = \begin{pmatrix} 0 & 1 \\ 1 & 0 \end{pmatrix} \quad (7)$$

are orbital operators build on wave functions (1) and (2), $Q_{\theta n}$ and $Q_{\varepsilon n}$ are cooperative symmetrized e_g distortions of oxygen ions in n th MnO₆ octahedron (the calculation and visualization is shown in Appendix), and V_e is a parameter of linear vibronic interaction.

In lanthanum manganite, the linear vibronic interaction was sufficient for the formation of orbital structure.²⁴ The orbital structure could be calculated, using the symmetrized distortion of nearest-neighbor oxygen coordination:

$$\cos \Phi_n^0 = \frac{Q_{\theta n}}{\rho_n}; \quad \sin \Phi_n^0 = \frac{Q_{\varepsilon n}}{\rho_n}; \quad \rho_n = \sqrt{Q_{\theta n}^2 + Q_{\varepsilon n}^2}. \quad (8)$$

In the strongly distorted BiMnO₃ crystal, the other types of vibronic interaction could affect the orbital structure. We suppose that the t_{1g} -type distortions of oxygen octahedra may change the proportion of $|\theta\rangle_n$, $|\varepsilon\rangle_n$ wave functions as in lanthanum titanate.²³ Also, the anharmonic interaction due to e_g distortions plays a role in the orbital state,²⁶

$$H_{QQ} = V_b \sum_n [(2Q_{zn}^2 - Q_{xn}^2 - Q_{yn}^2) X_{\theta n} + \sqrt{3}(Q_{xn}^2 - Q_{yn}^2) X_{\varepsilon n}] + N_e \sum_n [(Q_{\varepsilon n}^2 - Q_{\theta n}^2) X_{\theta n} + 2Q_{\theta n} Q_{\varepsilon n} X_{\varepsilon n}], \quad (9)$$

where $Q_{\theta n}$ and $Q_{\varepsilon n}$ are cooperative symmetrized e_g distortions of oxygen ions in n th MnO_6 octahedron, Q_{xn} , Q_{yn} , and Q_{zn} are cooperative symmetrized t_{1g} distortions of oxygen ions in n th MnO_6 octahedron (the calculation and visualization is shown in Appendix), V_b and N_e are vibronic constants.

The Bi^{3+} ions are strongly polarized and shifted from the symmetric position. A Bi^{3+} dipole moment affects the orbital state of Mn^{3+} ion. This effect could be taken into account in the vibronic constant. The bismuth sublattice symmetrized shifts are taken into account explicitly. Thus the effect of second-neighbor coordination of manganese ions must be taken into account like in Ref. 23,

$$H_R = V_e^R \sum_n (Q_{\theta n}^R X_{\theta n} + Q_{\varepsilon n}^R X_{\varepsilon n}), \quad (10)$$

where $Q_{\theta n}^R$ and $Q_{\varepsilon n}^R$ are cooperative symmetrized e_g distortions of bismuth ions of n th manganese ion (the calculation and visualization is shown in Appendix), V_e^R is the second-neighbor vibronic constant.

The contributions of Eqs. (9) and (10) must be smaller than that of Eq. (7), and they could change the general form of the ground state (1). The changes in Φ_n angles could be described as

$$\cos \Phi_n = \frac{\tilde{Q}_{\theta n}}{\tilde{\rho}_n}, \quad \sin \Phi_n = \frac{\tilde{Q}_{\varepsilon n}}{\tilde{\rho}_n}, \quad (11)$$

where

$$\begin{aligned} \tilde{Q}_{\theta n} = & Q_{\theta n} + \frac{V_b}{V_e} (2Q_{zn}^2 - Q_{xn}^2 - Q_{yn}^2) \\ & + \frac{N_e}{V_e} (Q_{\varepsilon n}^2 - Q_{\theta n}^2) + \frac{V_e^R}{V_e} Q_{\theta n}^R, \end{aligned} \quad (12)$$

$$\tilde{Q}_{\varepsilon n} = Q_{\varepsilon n} + \frac{\sqrt{3}V_b}{V_e} (Q_{xn}^2 - Q_{yn}^2) + \frac{2N_e}{V_e} Q_{\theta n} Q_{\varepsilon n} + \frac{V_e^R}{V_e} Q_{\varepsilon n}^R, \quad (13)$$

$$\tilde{\rho}_n = \sqrt{\tilde{Q}_{\theta n}^2 + \tilde{Q}_{\varepsilon n}^2}. \quad (14)$$

The constants of vibronic interaction in Eqs. (5)–(10) are not yet defined, but we can discuss their influence on the orbital structure of BiMnO_3 . Using the crystal structure data of Ref. 11, one can find the symmetrized distortions of oxygen and bismuth coordination of each manganese ion. The values of distortions are shown in Table II. There are four types of manganese ions distinguished by values and signs of symmetrized distortions. They are numbered in Fig. 1. The details of symmetrized distortion types and calculations are given in Appendix.

The orbital structure within the framework of linear vibronic interactions, presented by Eqs. (6)–(8), is

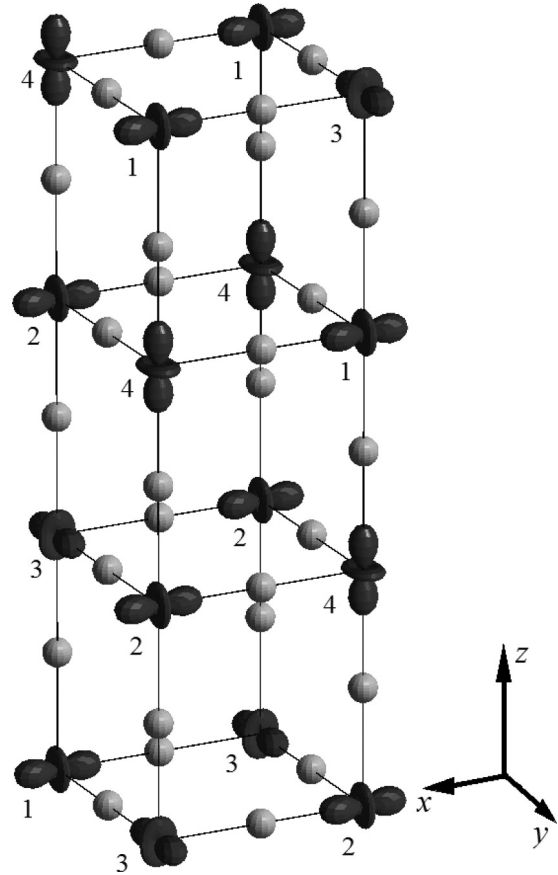


FIG. 1. Orbital structure of BiMnO_3 [approximated by Eqs. (6)–(8)]. The oxygen ions are denoted as spheres, the manganese ions are drawn as orbital distributions. The pseudoperovskite reference frame is drawn. For perception convenience, the deflected vertical Mn–Mn bonding line coincides with pseudoperovskite z axis. The true relation between pseudoperovskite z axis and Mn–Mn bond line see on Fig. 5 in Appendix.

approximately drawn in Fig. 1. The numerical values of orbital-mixing angles Φ_n are given in Table II.

The orbital structure of BiMnO_3 is considerably different when compared with the lanthanum manganite case.²⁴ The orbital mixing angles differ at various ions not only by the sign but also by value. This kind of orbital structure leads to qualitatively other types of magnetic ordering.

The influence of terms (8) and (9) upon the orbital structure will be discussed in another part of this paper. Because the calculations of V_b/V_e , N_e/V_e , and V_e^R/V_e relations for BiMnO_3 are not yet carried out, one needs some reasonable considerations to evaluate them in order to find the exact orbital structure. As a topic for discussion one could take a magnetic structure of the compound.

TABLE II. Orbital structure mixing angles of symmetrized distortions.

Position and number of Mn^{3+}	$Q_{\varepsilon n}$, Å	$Q_{\theta n}$, Å	Q_{xn} , Å	Q_{yn} , Å	Q_{zn} , Å	$Q_{\theta n}^R$, Å	$Q_{\varepsilon n}^R$, Å	Φ_n^0 , deg
4e, 1	0.26	−0.15	0.14	0.73	−0.58	−0.11	0.12	119
4e, 2	0.26	−0.15	0.14	0.73	−0.58	−0.11	0.12	120
4d, 3	−0.37	−0.16	−0.43	−1.02	0.28	−0.03	0.28	−113
4d, 4	−0.06	0.41	0.75	−0.16	1.14	−0.17	0.03	−8

III. SUPEREXCHANGE INTERACTION AND FRUSTRATED MAGNETIC STRUCTURE

Despite the complicated orbital structure, the magnetic ordering is of the simple ferromagnetic arrangement according to experimental investigations of Refs. 6, 8–11. The superexchange interaction could be described, using the orbital operators approach.²⁵ The Hamiltonian of superexchange interaction within Mn_2O_{11} cluster arranged along the pseudoperovskite z direction of undistorted crystal is

$$\hat{H}_{\text{ex}}^z = \frac{J_0 \cos^2 \varphi_{nm}}{r_{nm}^{10}} [1 + \alpha(X_{n\theta} + X_{m\theta}) + \beta X_{n\theta} X_{m\theta}] (\mathbf{S}_n \cdot \mathbf{S}_m), \quad (15)$$

where $X_{\theta n}$ and $X_{\theta m}$ are the orbital operators (7), $J_0 = 1.69 \times 10^4 \text{ K}\text{\AA}^{10}$, $\alpha = 1.0$, $\beta = 4.5$ from Ref. 25, r_{nm} is the average Mn–O distance in the interacting pair of ions n and m , φ_{nm} is the bonding angle Mn–O–Mn from the experimental studies of Ref. 11. If the Mn_2O_{11} cluster is arranged along the pseudoperovskite x or y direction, the superexchange interaction is²⁵

$$\hat{H}_{\text{ex}}^{x(y)} = \frac{J_0 \cos^2 \varphi_{nm}}{r_{nm}^{10}} \left[1 - \frac{\alpha}{2} (X_{n\theta} \pm \sqrt{3} X_{n\varepsilon} + X_{m\theta} \pm \sqrt{3} X_{m\varepsilon}) + \frac{\beta}{4} (X_{n\theta} \pm \sqrt{3} X_{n\varepsilon})(X_{m\theta} \pm \sqrt{3} X_{m\varepsilon}) \right] (\mathbf{S}_n \cdot \mathbf{S}_m). \quad (16)$$

Averaging the values of Hamiltonians (15) and (16), calculated on wave function (1), leads to a general superexchange interaction Hamiltonian

$$H_{\text{ex}} = \sum_{\alpha, nm} J_{\alpha}^{nm} (\Phi_n; \Phi_m) (\mathbf{S}_n \cdot \mathbf{S}_m), \quad (17)$$

where the exchange parameters could be found as functions of orbital mixing angles of each pair Φ_n , Φ_m , directed along pseudoperovskite axes $\alpha = x, y, z$:

$$J_{\alpha}^{nm} (\Phi_n, \Phi_m) = \frac{J_0 \cos^2 \varphi_{nm}}{r_{nm}^{10}} F_{\alpha} (\Phi_n, \Phi_m) \quad (18)$$

and

$$\begin{aligned} F_z (\Phi_n, \Phi_m) &= 1 + \alpha (\cos \Phi_n + \cos \Phi_m) + \beta \cos \Phi_n \cos \Phi_m, \\ F_{x,y} (\Phi_n, \Phi_m) &= 1 - \frac{\alpha}{2} (\cos \Phi_n \pm \sqrt{3} \sin \Phi_n + \cos \Phi_m \\ &\quad \pm \sqrt{3} \sin \Phi_m) + \frac{\beta}{4} (\cos \Phi_n \pm \sqrt{3} \sin \Phi_n) \\ &\quad \times (\cos \Phi_m \pm \sqrt{3} \sin \Phi_m). \end{aligned} \quad (19)$$

Within the framework of the linear vibronic interaction model and using the experimental crystal structure from Ref. 11, the superexchange parameters are calculated. The results are presented in Table III.

The superexchange interactions divide the magnetic subsystem of BiMnO_3 crystal into Oyz planes of pseudoperovskite reference frame. Within these planes, one could find ferromagnetic zigzag chains, coupled antiferromagnetically, as shown in Fig. 2. The interplane superexchange interaction is ferromagnetic, but the chains in the neighboring planes are shifted along the y axis relative to each other. This shift, together with the antiferromagnetic interaction, leads to a competition between exchange interactions in different pairs.

TABLE III. The superexchange parameters of BiMnO_3 [model set by Eqs. (1), (6)–(8), (18), and (19)]. The negative parameters denote FM type of interaction and positive ones correspond to AFM interaction.

Pairs of Mn^{3+} ions	Superexchange interaction, K		
	J_x	J_y	J_z
1–3	–12	–9	13
1–4	–13	13	–8
2–3	–12	–9	13
2–4	–12	13	–8

There are some magnetically frustrated cells with xy or xz orientation, which have three ferromagnetic bonds and one antiferromagnetic one. Taking into account the near values of different kinds of superexchange interaction, we come to a conclusion that the magnetic structure could not be simply determined as ferromagnetic. There are three types of magnetic structures that could be formed in the crystal with similar orbital structures like on Fig. 1 and superexchange signs like on Fig. 2: *F*-type ferromagnetic, *A*-type antiferromagnetic (with FM Oxy planes), and *C*-type antiferromagnetic (with FM Ox chains).

The exchange parameters shown in Table III within the framework of linear nearest-neighbor vibronic interaction (6) describe the AFM *A*-type structure. A variation of phenomenological J_0 , α , and β constants of orbitally dependent exchange interaction (18) and (19) within the range of 20% near values of Ref. 25 is not enough to shift the magnetic ground state from AFM order to FM one. Thus we conclude that the vibronic Hamiltonian must be more complicated than it is supposed in Eq. (5).

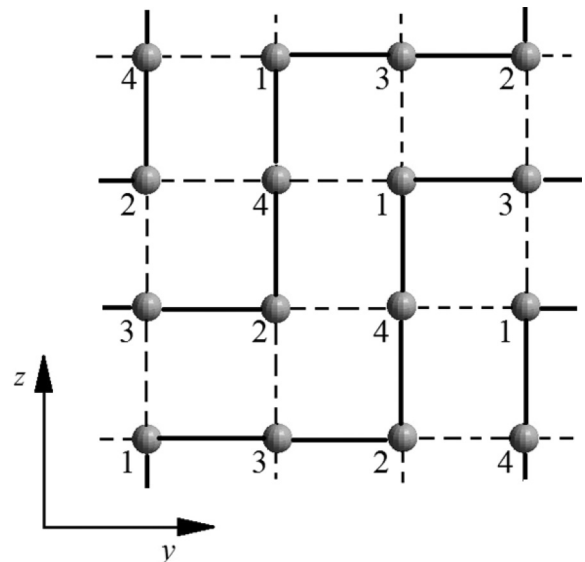
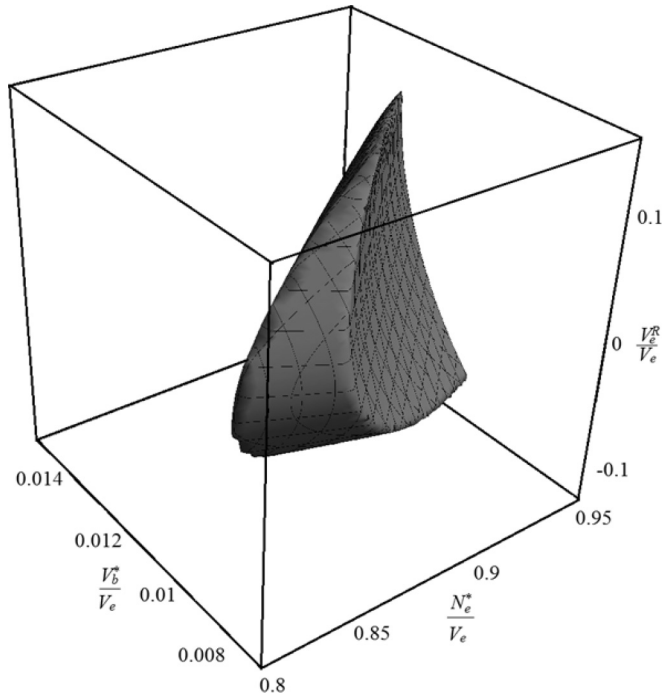


FIG. 2. Ferromagnetic zigzag chains (solid lines) in pseudoperovskite Oyz plane. Dashed lines denote antiferromagnetic bonds. Spheres denote magnetic Mn^{3+} ions.


 FIG. 3. The diagram of V_b^*/V_e , N_e^*/V_e , and V_e^R/V_e relations.

IV. RESULTS AND DISCUSSION

In order to determine a possibility of FM structure depending upon V_b/V_e , N_e/V_e , and V_e^R/V_e relations, the following considerations must be taken into account: (1) the experimental value of Curie temperature is $T_C = 100$ K.¹¹ The superexchange parameters must give an adequate estimation of Curie temperature. (2) The energies of *C*- and *A*-type antiferromagnetic structures must be higher than the *F*-type one. (3) The terms of additional contributions in orbital structure must be at least at a lower absolute value than the main term due to linear vibronic interaction.

The comparison of constants having different units is inconvenient. In order to bring V_b/V_e , N_e/V_e relations in dimensionless form, the N_e and V_b could be multiplied by Bohr radius $a_0 = 0.529$ Å. This value is typical for ionic displacements and is of the same order with symmetrized distortions in the crystal. New constants $V_b^* = V_b a_0$ and $N_e^* = N_e a_0$ produce dimensionless relations V_b^*/V_e , N_e^*/V_e .

In Fig. 3, the region of V_b^*/V_e , N_e^*/V_e , and V_e^R/V_e relations satisfying conditions 1–3 is drawn. The Curie temperature is taken approximately $T_C \approx 100 \pm 5$ K within

the framework of the mean-field approximation. The diagram shows that the crucial role of forming ferromagnetism belongs to the anharmonic term of e_g distortions [the second summand of Eq. (9)]. The N_e constant must be sufficiently large. The modified orbital structure generally remains of the same form as in Fig. 1. The forms of orbitals are noticeably changed only in manganese ions of third position from $d_{3y^2-r^2}$ to $d_{y^2-z^2}$.

The exchange parameters, being approximately of the same magnitude according to Table III, became more anisotropic. The largest values belong to the J_x parameters (≥ 25 K), and the interactions within the *Oyz* plane must be weaker (~ 6 K). The basic feature of the interactions is present, i.e., the signs of superexchange parameters remain the same. Examples of possible structures and the corresponding set of superexchange parameters are shown in Table IV and Fig. 4.

The shown orbital structure is the closest to the Jahn-Teller one, but it is not the only structure determined by the diagram in Fig. 3. The marginal regions of vibronic constants diagrammed in Fig. 3 provide the drastic change of superexchange parameters relation, which could not be taken into account.

The values of vibronic constants we could compare with the *ab initio* calculations of Ref. 23 for LaTiO_3 : $V_b^*/V_e = -0.02$, $V_e^R/V_e = 0.15$. The values in our estimation are of the same order. The value of N_e constant is the largest among the additional vibronic interactions. It could be compared with earlier MO LKAO calculations of the same interaction parameters in KCuF_3 crystal²⁶ where the relation is $N_e^*/V_e = 0.2$ and with RbMnF_3 ²⁷ where for an excited manganese state the relation is $N_e^*/V_e = 0.73$. The comparison shows that the large value used for the calculation in the example in Table III, $N_e^*/V_e = 0.82$, could take place in the case of manganese ions. Thus the adjustment of orbital structure changes the parameters of superexchange interaction twice in the most pairs. The FM interaction in *Ox* pseudoperovskite direction plays the main role in FM ordering.

The values of superexchange interaction were calculated by Soloviev *et al.*^{7,18} They found the magnitude of FM interaction as $|J_{NN}| \sim 60$ K. It is larger than our values and does not provide the Curie temperature of the experimental value $T_C = 100$ K. The antiferromagnetic interaction parameters of Refs. 7 and 18 are of the same order. These investigations use the next-nearest neighbor antiferromagnetic exchange interaction to explain the FE state via a canted magnetic structure, but there are no experimental studies of single crystals confirming FE properties of BiMnO_3 . Thus using the long-range exchange interaction of the large values (~ 10 K) seems to be not sufficiently argued.

 TABLE IV. Example of possible orbital structure of BiMnO_3 due to anharmonic and next-nearest neighbor vibronic interactions [Eqs. (9)–(14)].

Vibronic interaction parameters	Orbital structure angles, degrees	Pair of Mn^{3+} ions	Superexchange interaction, K		
			J_x	J_y	J_z
$V_b^*/V_e = 0.008$	$\Phi_1 = 120$	1–3	–24	–7	7
$N_e^*/V_e = 0.82$	$\Phi_2 = 120$	1–4	–26	6	–7
$V_e^R/V_e = 0.05$	$\Phi_3 = -92$	2–3	–25	–6	6
	$\Phi_4 = -35$	2–4	–24	6	–7

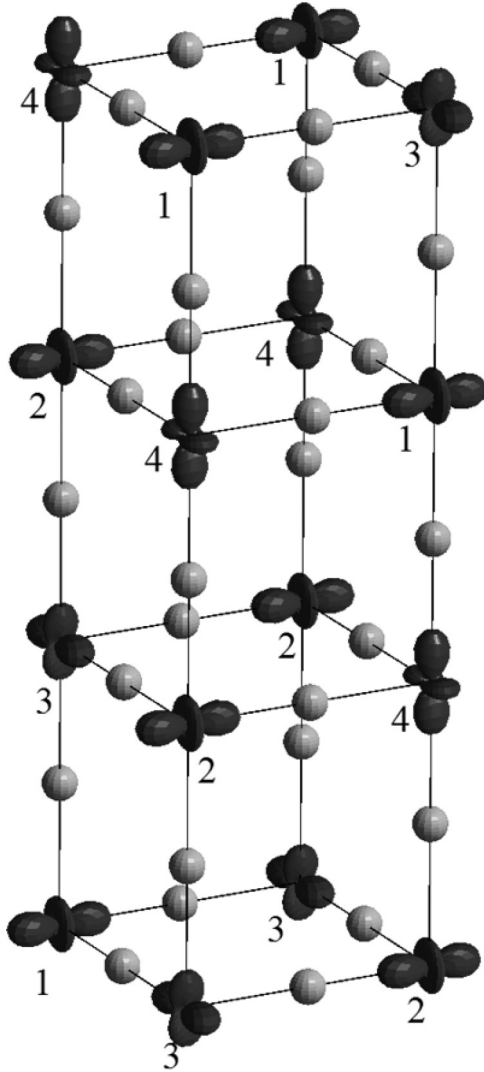


FIG. 4. The example of orbital structure giving FM ordering.

In Fig. 2, one could recognize the same zigzag ordering of CO manganite $\text{La}_{0.5}\text{Ca}_{0.5}\text{MnO}_3$.²⁴ However, CE magnetic structure like in CO manganites is impossible in this case, because the pseudoperovskite Oxy planes with FM interaction between each other are ordered with a shift. This shift causes a competition between superexchange interactions.

The reasons explaining the modified vibronic interaction could be used for the explanation of magnetic ordering in the frustrated CO manganite $\text{La}_{0.333}\text{Ca}_{0.667}\text{MnO}_3$.²⁸ The orbital structure due to linear vibronic interaction must be corrected by changing the orbital mixing angle at 5° in order to describe the magnetic unit cell. The reason for the correction could be the additional terms of vibronic Hamiltonian.

The effect of anharmonicity in regular manganites $R\text{MnO}_3$ ($R = \text{La}, \text{Nd}, \text{Pr}$) also could be taken into account, but the adjusted orbital structure produces no change in signs of superexchange interaction parameters. Thus the magnetic structure without frustration remains the same in spite of the difference between the adjusted orbital structure and ordering due to linear vibronic interaction.

V. CONCLUSIONS

Thus the advanced model of the vibronic interaction in Jahn-Teller crystals is proposed. This model could be applied to orbital structure description in the case of magnetic frustration, in order to explain the complicated magnetic structures. The necessity to expand the vibronic Hamiltonian of magnetically frustrated compounds is proved by the fact that the weak interaction changing orbital structure affects the superexchange interaction equilibrium, moving it to another type of magnetic structure. In the case of bismuth manganite, the equilibrium antiferromagnetic state changes to a ferromagnetic one due to the orbital anharmonicity and next-nearest neighbor vibronic interaction. The Curie temperature of BiMnO_3 is also explained by this model.

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APPENDIX: SYMMETRIZED DISTORTION CALCULATIONS

The main problem of the symmetrized distortion calculation is the undistorted axes choice. In the lanthanum manganite^{24,28} and titanate²³ crystals, there were the MO_6 octahedra ($M = \text{Mn}^{3+}, \text{Ti}^{3+}$ JT ions) with orthogonal $M\text{-O}$ axes. Thus one could make a projection on the undistorted axes taking into account the octahedron rotation. In BiMnO_3 case, there is a non-Cartesian local reference frame, associated with MnO_6 octahedron axes.

In order to choose undistorted octahedron axes, we assume pseudoperovskite x and y axes directed along orthogonal Mn1-Mn3 bonds, and the z axis is chosen to be directed perpendicular to Oxy plane along one of Mn2-Mn4 bonds. The Mn1-Mn3 nonorthogonal bond is deflected from this axis (like z axis in Fig. 2). In neighboring octahedron, the Mn2-Mn4 bond is deflected from x axis. The position coordinates and the undistorted octahedron axes have to be transferred to the Cartesian reference frame. The orthonormal basis of this reference frame, associated with monoclinic unit cell, are

$$\mathbf{i}_m = \frac{\mathbf{a}}{a}, \quad \mathbf{j}_m = \frac{\mathbf{b}}{b}, \quad \mathbf{k}_m = \frac{1}{\sin \beta} \left(\frac{\mathbf{c}}{c} - \frac{\mathbf{a}}{a} \cos \beta \right), \quad (\text{A1})$$

where a, b, c are lattice constants of BiMnO_3 and $\mathbf{a}, \mathbf{b}, \mathbf{c}$ are vectors of BiMnO_3 unit cell.

Within the reference frame (A1), the undistorted octahedron axes are directed along the perovskite orthonormal basis:

$$\begin{aligned} \mathbf{i} &= (-0.82; -0.04; 0.58), \\ \mathbf{j} &= (-0.37; 0.73; -0.57), \\ \mathbf{k} &= (-0.40; 0.68; 0.61). \end{aligned} \quad (\text{A2})$$

A correlation between octahedron axes is drawn in Fig. 5. The types and expressions of symmetrized distortions coordinates are compiled from Ref. 23.

The e_g -type symmetrized distortions of each MnO_6 octahedron are drawn in Fig. 6 (number n of manganese ion is

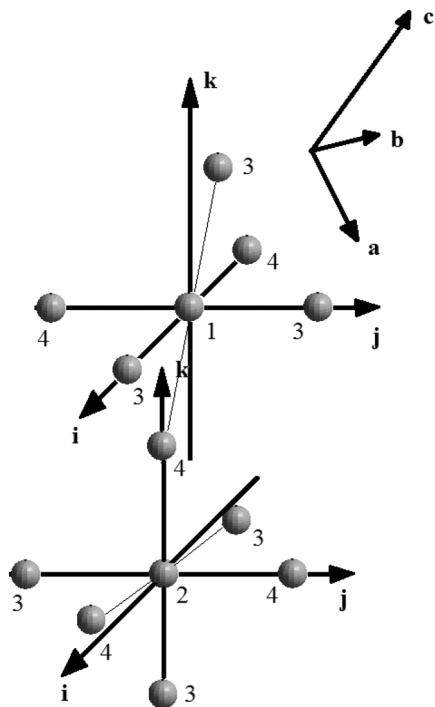


FIG. 5. Pseudoperovskite reference (i, j, k) frame compared with monoklinic reference frame (a, b, c). Numbers denote manganese position according to Table II.

omitted):

$$Q_\theta = [- (\mathbf{r}_{\text{MnO}_1} - \mathbf{r}_1) \cdot \mathbf{i} - (\mathbf{r}_{\text{MnO}_2} - \mathbf{r}_2) \cdot \mathbf{j} + 2(\mathbf{r}_{\text{MnO}_3} - \mathbf{r}_3) \cdot \mathbf{k} + (\mathbf{r}_{\text{MnO}_5} - \mathbf{r}_5) \cdot \mathbf{i} + (\mathbf{r}_{\text{MnO}_4} - \mathbf{r}_4) \cdot \mathbf{j} - 2(\mathbf{r}_{\text{MnO}_6} - \mathbf{r}_6) \cdot \mathbf{k}] / \sqrt{12},$$

$$Q_\varepsilon = [(\mathbf{r}_{\text{MnO}_1} - \mathbf{r}_1) \cdot \mathbf{i} - (\mathbf{r}_{\text{MnO}_2} - \mathbf{r}_2) \cdot \mathbf{j} - (\mathbf{r}_{\text{MnO}_5} - \mathbf{r}_5) \cdot \mathbf{i} + (\mathbf{r}_{\text{MnO}_4} - \mathbf{r}_4) \cdot \mathbf{j}] / 2, \quad (\text{A3})$$

where \mathbf{r}_i ($i = 1, \dots, 6$) are Mn–O distances of undistorted octahedron, numbered according to Fig. 6,

$$\begin{aligned} \mathbf{r}_1 &= r_0 \mathbf{i}, & \mathbf{r}_2 &= r_0 \mathbf{j}, & \mathbf{r}_3 &= r_0 \mathbf{k}, \\ \mathbf{r}_4 &= -r_0 \mathbf{j}, & \mathbf{r}_5 &= -r_0 \mathbf{i}, & \mathbf{r}_6 &= r_0 \mathbf{k}, \end{aligned} \quad (\text{A4})$$

$$r_0 = \frac{1}{6} \sum_{i=1}^6 |\mathbf{r}_{\text{MnO}_i}|,$$

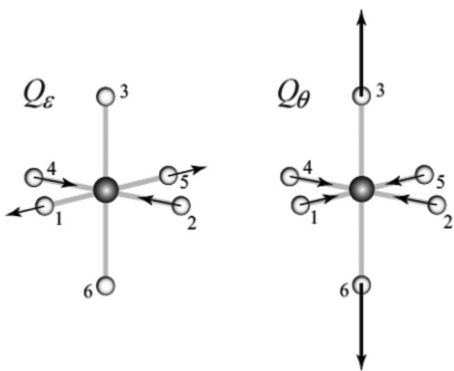


FIG. 6. Symmetrized e_g distortion of MnO_6 octahedron. Dark spheres denote manganese ions, light spheres denote oxygen ions, numbers indicate different oxygen positions.

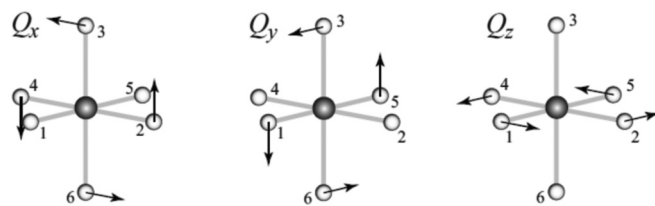


FIG. 7. Symmetrized t_{1g} distortion of MnO_6 octahedron. Dark spheres denote manganese ions, light spheres denote oxygen ions, numbers indicate different oxygen positions.

and $\mathbf{r}_{\text{MnO}_i}$ ($i = 1, \dots, 6$) are the Mn–O distances of distorted octahedron in the crystal.

The t_{1g} -type symmetrized distortions of each MnO_6 octahedron are drawn in Fig. 7 (number n of manganese ion is omitted),

$$\begin{aligned} Q_x &= [(\mathbf{r}_{\text{MnO}_2} - \mathbf{r}_2) \cdot \mathbf{k} - (\mathbf{r}_{\text{MnO}_4} - \mathbf{r}_4) \cdot \mathbf{k} - (\mathbf{r}_{\text{MnO}_3} - \mathbf{r}_3) \cdot \mathbf{j} + (\mathbf{r}_{\text{MnO}_6} - \mathbf{r}_6) \cdot \mathbf{j}] / 2, \\ Q_y &= [-(\mathbf{r}_{\text{MnO}_1} - \mathbf{r}_1) \cdot \mathbf{k} + (\mathbf{r}_{\text{MnO}_5} - \mathbf{r}_5) \cdot \mathbf{k} - (\mathbf{r}_{\text{MnO}_3} - \mathbf{r}_3) \cdot \mathbf{i} - (\mathbf{r}_{\text{MnO}_6} - \mathbf{r}_6) \cdot \mathbf{i}] / 2, \\ Q_z &= [(\mathbf{r}_{\text{MnO}_1} - \mathbf{r}_1) \cdot \mathbf{j} - (\mathbf{r}_{\text{MnO}_5} - \mathbf{r}_5) \cdot \mathbf{j} - (\mathbf{r}_{\text{MnO}_2} - \mathbf{r}_2) \cdot \mathbf{i} + (\mathbf{r}_{\text{MnO}_4} - \mathbf{r}_4) \cdot \mathbf{i}] / 2. \end{aligned} \quad (\text{A5})$$

The e_g -type symmetrized distortions of next-nearest neighbor. The bismuth coordination of manganese ion is drawn in Fig. 8 (number n is omitted);

$$\begin{aligned} Q_\varepsilon^R &= [(\mathbf{r}_{\text{MnBi}_1} - \mathbf{r}_1) \cdot (-\mathbf{i} + \mathbf{j}) + (\mathbf{r}_{\text{MnBi}_2} - \mathbf{r}_2) \cdot (\mathbf{i} + \mathbf{j}) + (\mathbf{r}_{\text{MnBi}_3} - \mathbf{r}_3) \cdot (\mathbf{i} - \mathbf{j}) + (\mathbf{r}_{\text{MnBi}_4} - \mathbf{r}_4) \cdot (-\mathbf{i} - \mathbf{j}) + (\mathbf{r}_{\text{MnBi}_5} - \mathbf{r}_5) \cdot (-\mathbf{i} + \mathbf{j}) + (\mathbf{r}_{\text{MnBi}_6} - \mathbf{r}_6) \cdot (\mathbf{i} + \mathbf{j}) + (\mathbf{r}_{\text{MnBi}_7} - \mathbf{r}_7) \cdot (\mathbf{i} - \mathbf{j}) + (\mathbf{r}_{\text{MnBi}_8} - \mathbf{r}_8) \cdot (-\mathbf{i} - \mathbf{j})] / 4, \\ Q_\theta^R &= [(\mathbf{r}_{\text{MnBi}_1} - \mathbf{r}_1) \cdot (-\mathbf{i} - \mathbf{j} + 2\mathbf{k}) + (\mathbf{r}_{\text{MnBi}_2} - \mathbf{r}_2) \cdot (\mathbf{i} - \mathbf{j} + 2\mathbf{k}) + (\mathbf{r}_{\text{MnBi}_3} - \mathbf{r}_3) \cdot (\mathbf{i} + \mathbf{j} + 2\mathbf{k}) + (\mathbf{r}_{\text{MnBi}_4} - \mathbf{r}_4) \cdot (-\mathbf{i} + \mathbf{j} + 2\mathbf{k}) + (\mathbf{r}_{\text{MnBi}_5} - \mathbf{r}_5) \cdot (-\mathbf{i} - \mathbf{j} - 2\mathbf{k}) + (\mathbf{r}_{\text{MnBi}_6} - \mathbf{r}_6) \cdot (\mathbf{i} - \mathbf{j} - 2\mathbf{k})] \end{aligned}$$

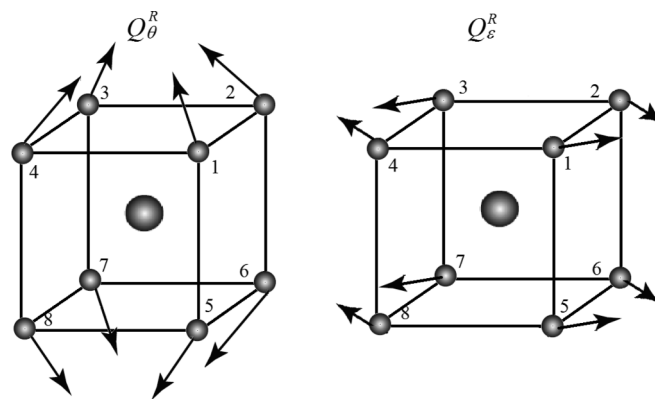


FIG. 8. Symmetrized e_g distortion of MnBi_8 cluster. Large spheres denote manganese ions, small spheres denote bismuth ions, numbers indicate different bismuth positions.

$$+ (\mathbf{r}_{\text{MnBi}_7} - \mathbf{r}_7) \cdot (\mathbf{i} + \mathbf{j} - 2\mathbf{k}) \\ + (\mathbf{r}_{\text{MnBi}_8} - \mathbf{r}_8) \cdot (-\mathbf{i} + \mathbf{j} - 2\mathbf{k})] / (2\sqrt{12}), \quad (\text{A6})$$

where \mathbf{r}_i ($i = 1, \dots, 8$) are Mn–Bi distances of bismuth environment, numbered according to Fig. 8,

$$\mathbf{r}_1 = R_0(\mathbf{i} + \mathbf{j} + \mathbf{k})/\sqrt{3}, \\ \mathbf{r}_2 = R_0(-\mathbf{i} + \mathbf{j} + \mathbf{k})/\sqrt{3}, \\ \mathbf{r}_3 = R_0(-\mathbf{i} - \mathbf{j} + \mathbf{k})/\sqrt{3}, \\ \mathbf{r}_4 = R_0(\mathbf{i} - \mathbf{j} + \mathbf{k})/\sqrt{3},$$

$$\mathbf{r}_5 = R_0(\mathbf{i} + \mathbf{j} - \mathbf{k})/\sqrt{3}, \\ \mathbf{r}_6 = R_0(-\mathbf{i} + \mathbf{j} - \mathbf{k})/\sqrt{3}, \\ \mathbf{r}_7 = R_0(-\mathbf{i} - \mathbf{j} - \mathbf{k})/\sqrt{3}, \\ \mathbf{r}_8 = R_0(\mathbf{i} - \mathbf{j} - \mathbf{k})/\sqrt{3}, \\ R_0 = \frac{1}{8} \sum_{i=1}^8 |\mathbf{r}_{\text{MnBi}_i}|, \quad (\text{A7})$$

and $\mathbf{r}_{\text{MnBi}_i}$ ($i = 1, \dots, 8$) are Mn–Bi distances of distorted crystal.

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